Tuning Structural and Optical Properties of WO₃ NPs Thin Films by the Fluency of Laser Pulses

Ali Jaafar Hwaidi^{1,2}, Nadheer Jassim Mohammed^{1*}

¹Thin film laboratory, Department of Physics, College of Science, Mustansiriyah University, Baghdad, IRAQ. ²Ministry of Education, Directorate General of Education Rusafa 2, Baghdad, IRAQ.

*Correspondent contact: <u>nadheerphys@uomustansiriyah.edu.iq</u>

Article Info	ABSTRACT
Received 21/03/2022	In this paper, tungsten oxide thin films were synthesized successfully by the laser pulse deposition (PLD) method using a pulsed laser (ND-YAG) and wavelength (1064 nm) on a glass substrate at different laser fluencies. The effect of increasing laser fluency on the optical and structural properties of WO3 nanoparticle thin films were investigated by UV-Visible spectrophotometer,
Accepted 24/04/2022	X-Ray diffraction (XRD), atomic force microscope (AFM) and Scanning Electron Microscope (SEM). X-Ray measurements for all samples of WO ₃ NPs thin films have shown that by increasing the laser fluencies from 5.175 to 6.369 J/cm ² , the intensity of the (2 01) diffraction peak increases due to the film continuing to grow with increased crystallization.
Published 25/09/2022	KEYWORDS : WO ₃ ; Thin films; pulsed laser deposition; vacuum; nanoparticles.

INTRODUCTION

One of the semiconductor metal oxides is tungsten oxide WO₃, which has an energy gap ranging from 2.6-3.0 eV, due to its unique features, including electrochromism, photochromism, gas sensing, thermoelectric, and catalytic capabilities; it has been widely applications [1-7]. Smart windows, electronic information displays, electrochromic sensors and photocatalysts, devices. gas photovoltaic devices, and photo-electrochemical devices are all possible uses. [8-12]. Tungsten oxide has lately been used as an efficient hole (for WO₃) and electron (for WO₃ X) injection/transport layer in hybrid organic-inorganic optoelectronic devices due to its high work function of roughly 6.2 eV and great transparency in the visible region [13, 14]. Sintering the $W_{18}O_{49}$ nanorod film yielded WO₃ nanorods. which as were used photoelectrodes in dye-sensitized solar cells [15]. Ramgir et al., have revealed that modifying WO₃ films with CuO improves their H₂S sensing characteristics. [16]. Because of numerous configurations, such as triclinic [17], orthorhombic [18], monoclinic [19], and oxygen deficient tungsten oxide structural changes, tungsten oxide is a difficult material in terms of crystal structure and thermal stability [20]. The typical WO₃ crystal structure is a cubic ReO₃ structure with an octahedral alignment of oxygen atoms enclosing the action, and the polymorphs of WO3 are aberrations from the cubic structure [21]. As a result, the WO₃ structure is comprised of a threedimensional lattice of WO₆ octahedral corner For various applications, sharing. unique compositions and structures of WO₃ coatings are commonly preferred. Amorphous WO₃ films with coloring efficiency high and auick coloration/bleaching kinetics are commonly employed in monitors and color memory chips, but polycrystalline WO₃ films exhibiting strong gas sensors sensitivities can be used as ambient gas sensors. Zhang et al. developed a WO₃ reactive film-based gas sensor with a monoclinic phase structure that demonstrated a greater sensitivity and much more selective identification of NO2 at ambient temperature when illuminated by visible light. [22]. Diverse deposition processes, like as chemical vapor deposition, have already been established thus far to create WO3 films with distinct properties, crystallinity and structure [23, 24], preparation by laser, spray coating, electrosynthesis, spin deposition, sol-gel techniques, flashing, heat deposition and degradation of WO₃ layers [25-37]. Within these techniques, pulse laser



94



deposition has indeed been largely regarded for the expansion of handset films, and it is encouraging in depositing WO3 layers over traditional coating methods number of advantages, such as low preparation temperature, better wet to the substrate, reversibility and predictability of stoichiometry and crystal composition, and the simple synthesis of composite materials and mixtures of materials with various vapor pressures. As a result, the PLD manufacturing method is well suited for electrochromism technologies based on the synthesized WO₃ electrochromic layer. The synthesizing processes and deposition circumstances, such as substrate temperature and fluence density, have a significant impact on the microstructure, constitution, shape, and physical characteristics of WO₃ thin films by laser pulses. The current research focuses on the depositing of WO₃ layers on glass substrates using laser pulses, as well as a comprehensive examination of the impact of various laser energies on the structural, topological, and spectroscopy aspects of the formed WO₃ layers.

EXPERIMENTAL PROCEDURE

In this work, WO₃ thin films were fabricated under different conditions using powder with purity of 99.99%. The powder was pressed as pellet with 1.5 and 0.3 cm diameter and thickness, respectively. At a pressure of 7 ton, the hydraulic position was employed for 15 minutes. The tungsten oxide layer was deposited on glass substrates $(1.5 \times 1.5 \text{ cm})$. These glass substrates were cleaned with distilled water to eliminate any leftover dust and filth from their surfaces. After that, an ultrasonic device was used to wash the glass substrates in alcohol for 5 minutes to remove certain oxides and grease. The air from shoveling was utilized to dry the glass substrates in this technique. Lastly, light paper has been used to wash the plates before using a Nd:Yag laser to prepare films at room temperature using the laser pulses technique with wavelengths of 1064 nm and fluencies of 5.175, 5.573, 5.97, and 6.369 J/cm^2 . The repetition rate was 5Hz, and it took place at a 450 angle on the pellet surface. Under low pressure $(1 \times 10^{-3} \text{ mbar})$ and oxygen (1mbar), the spacing between substrate and WO₃ pellet is 1.5 cm [38], as seen in Figure 1.

RESULTS AND DISCUSSION Structural studies

The effect of increased laser fluencies on crystal structure and orientation of all the WO₃ NPs thin films through X-ray examinations can be observed. Since this increase has an effect that can be indexed to the polycrystalline structure compared to JCPDS card no. 00.32.1395 monoclinic (a=7.309, b= 7.522, c= 7.678), as shown in Figure 2.



Figure 1. The pulsed laser deposition process is depicted schematically [38].



Figure 2. X-ray diffraction of WO₃ NPs thin films at different laser fluencies 5.175, 5.573, 5.97 and 6.369 J/cm².

WO₃ NPs thin films have been showing that the sharp peak at $2\theta = 26.827$ correspond to ($\overline{2}01$) plane and exhibited preferential orientation along ($\overline{2}01$) plane of all the samples, as shown in Figure 3.



Figure 3. X-ray diffraction in the $(\overline{2}01)$ plane of WO3 NPs thin films at different laser fluencies 5.175, 5.573, 5.97 and 6.369 J/cm2.

The increase and decrease in the broadness of $(\overline{2}01)$ diffraction peak by increasing the laser fluencies from 5.175 to 5.573 J/cm² are due to the decrease in the crystal size from 85.09 to 64.7 nm and from 5.97 to 6.369 J/cm² due to the increase in the crystal size from 95.7 to 97.2 nm, respectively. By increasing the laser fluencies from 5.175 to 5.573 J/cm² the intensity of ($\overline{2}01$) diffraction peak increases due to the film continuing to grow with increased crystallization [39].

As the laser fluencies continue to increase from 5.573 to 6.369 J/cm², we notice a decrease in the intensity of the ($\overline{2}01$) diffraction peak due to the decreases in the thickness of WO₃ NPs thin films and decrease in the adsorbed atoms on the substrate surface [40].

SEM images show a clear effect of increasing the laser fluencies in the range 5.175, 5.573, 5.97 and 6.369 J/cm² on the prepared films through the interspaces between the nanoparticles, as shown in Figure 4. The increased laser fluencies provide sufficient heat for the particles to cause them to coalesce and aggregate, thus reducing the interfacial distances between them [41].

It is also evident from the SEM images that the particle size of WO_3 NPs decreased and increased as 37.96, 34.5, 67.26, and 87.9 nm with the increase in the laser fluency of 5.175, 5.573, 5.97, and 6.369, respectively, as shown in Figure 4, a; b; c; and d.

EDX measurements of the WO_3 NPs were accomplished to monitor the increase in the concentration of the ratio O/W, shown with the SEM images in the Figure 4.

As laser fluency increases, the ratio of O/W also increases as a result of oxidation processes.

• •

Whereas the increase in the percentage of oxygen indicates the atom's tendency to gain an electron to reach a stable state, as shown in Figure 5.

The three-dimensional AFM micrograph and granularity of the WO₃ NPs thin films as deposition. By investigating AFM, we note that the deposited thin films are in the form of small connected crystal granules distributed regularly without cracks, as shown in the Figure 6, a; b; c and d. As the laser fluencies were increased from 5.175 to 6.369 J/cm2, the roughness and RMS of WO3NPs thin films increased, as shown in Table 1.



Figure 4. SEM and EDS images with particle size of WO_3 NPs: a, b, c and d at the fluencies 5.175, 5.573, 5.97 and 6.369 J/cm², respectively.





Figure 5. Concentration of the ratio O/W of WO₃ NPs thin films at the fluencies 5.175, 5.573, 5.97 and 6.369 J/cm²cm).



Figure 6. AFM and granularity images of WO₃ NPs: a, b, c and d at the fluencies 5.175, 5.573, 5.97 and 6.369 J/cm², respectively.

Table 1. The laser fluencies with the values of Sa (roughness average) and Sq (root mean square).

Laser fluency (J/cm ²)	Sa (roughness average (nm)	RMS (root mean square (nm)
5.175	2	2.39
5.573	14.8	18
5.97	17.3	21.3
6.369	27	31.8

This is because the removal of atoms from the target increases as the fluencies of laser energy increase, resulting in more melting and fusing [42].

In addition, the fluencies of high laser energies improve the coalescence process of neighboring atoms in the crystal [41].

Optical studies

The absorbance intensities of WO_3 NPs thin films were recorded by spectrophotometer within the wavelength ranges 300-900 nm. The absorbance spectra of WO_3 NPs exhibit increased behavior with increasing all laser fluencies for the wavelengths above 400 nm, as shown in the Figure 7.

For the absorbance spectra in the range less than 400 nm, different behavior can be observed where all the absorbance values increase with increasing laser fluencies to reach the maximum values at wavelength 305 nm except for laser fluency 6.369 J/cm² which is at wavelength 315 nm to give the red shift, as shown in the Figure 8.



Figure 7. The absorbance spectra of WO₃NPs thin films at the fluencies 5.175, 5.573, 5.97 and 6.369 J/cm².



Figure 8. The absorbance spectra of WO₃NPs thin films at the fluencies 5.175, 5.573, 5.97 and 6.369 J/cm².

This behavior is probably ascribed to the increase of particle sizes and surface roughness with increasing laser fluencies [43]. For incident photon energy greater than the band gap and above the exponential, the optical absorption follows a power law [44]:

$$(\alpha h v) = \beta \left(h v - E_g \right)^n \tag{1}$$

Where hv is the incident photon energy, β is the edge with parameter and n is an exponent ,Eg is the optical bandgap, and α is the absorption coefficient.

Figure 9 depicts the direct optical energy gap of WO_3 NPs for various laser fluencies. When the laser fluencies were increased from 5.175 to 5.57 J/cm², the energy gap was raised because of increased local levels within the energy band gap due to enhanced recrystallization of the thin film and a change in the crystal phase that reduced crystallite size [45].

Continuously increasing the fluency of the laser from 5.57 to 6.36 J/cm² decreases the energy bandgap due to local reduction within the energy bandgap due to improved crystallization as well as increased crystal size [46], as shown in Figure 9.



Figure 9. Energy gap (E_g) of WO₃ NPs thin films at the fluencies 5.175, 5.573, 5.97 and 6.369 J/cm².

CONCLUSIONS

In this study, the mechanism of increasing the laser fluency made prepare thin films of WO₃ NPs with high crystalline at 5.573 J/cm² of the diffraction peak ($\overline{2}01$). This increase also leads to the stability of the prepared films as a result of oxidation processes by increasing the ratio O\W. As the laser fluencies were increased from 5.175 to 6.369 J/cm², the roughness and RMS of WO₃NPs thin films increased. This increases the possibility of employing the prepared films in the field of manufacturing gas sensors.

The increase of laser fluencies from 5.175 to 5.97 J/cm² led to reaching the values of absorbance maxima at wavelengths of 305 nm except for laser

fluency 6.369 J/cm² which is at wavelength 315 nm to give the red-shift. The absence of a shift in the maximum values of the absorbance at the fluencies from 5.175 to 5.573 J/cm² enables us to use the prepared films in the applications of photodetectors. Increasing the fluency of laser from 5.175 to 5.573 J/cm² lead to an increase in the optical energy bandgap this leads to blue shift. Otherwise increasing from 5.573 to 6.369 J/cm² gave the red shift.

ACKNOWLEDGMENT

This work was done with the possibilities available in the thin film laboratory of the Physics Department at the Faculty of Science, Mustansiriyah University, Baghdad, Iraq.

REFERENCES

[1] D. Zhang, Y. Cao, J. Wu and X. Zhang, Tungsten trioxide nanoparticles decorated tungsten disulfide nanoheterojunction for highly sensitive ethanol gas sensing application, Appl. Surf. Sci., 2020, 503, 144063.

https://doi.org/10.1016/j.apsusc.2019.144063

[2] X. Chang, S. Xu, S. Liu, N. Wang, S. Sun and X. Zhu, et al., Highly sensitive acetone sensor based on WO3 nanosheets derived from WS2 nanoparticles with inorganic fullerene-like structures, Sens. Actuators, B, 2021, 343, 130135.

https://doi.org/10.1016/j.snb.2021.130135

- [3] A. Staerz, S. Somacescu, M. Epifani, T. Kida, U. Weimar and N. Barsan, WO3-based gas sensors: identifying inherent qualities and understanding the sensing mechanism, ACS Sens., 2020, 5, 1624-1633. <u>https://doi.org/10.1021/acssensors.0c00113</u>
- [4] Q. Ding, Y. Wang, P. Guo, J. Li, C. Chen and T. Wang, et al., Cr-doped urchin-like WO3 hollow spheres: the cooperative modulation of crystal growth and energyband structure for high-sensitive acetone detection, Sensors, 2020, 20, 3473. <u>https://doi.org/10.3390/s20123473</u>
- [5] S. Liu, W. Zeng and Y. Li, Synthesis of spherical WO3·H2O network for ethanol sensing application, Mater. Lett., 2019, 253, 42-45. https://doi.org/10.1016/j.matlet.2019.06.037
- [6] C. Dong, R. Zhao, L. Yao, Y. Ran, X. Zhang and Y. Wang, A review on WO3 based gas sensors: morphology control and enhanced sensing properties, J. Alloys Compd., 2020, 820, 153194. https://doi.org/10.1016/j.jallcom.2019.153194
- [7] C.-H. Chang, T.-C. Chou, W.-C. Chen, J.-S. Niu, K.-W. Lin and S.-Y. Cheng, et al., Study of a WO3 thin film



98



based hydrogen gas sensor decorated with platinum nanoparticles, Sens. Actuators, B, 2020, 317, 128145. https://doi.org/10.1016/j.snb.2020.128145

- [8] H.J. Chen, N.S. Xu, S.Z. Deng, D.Y. Lu, Z.L. Li, J. Zhou, J. Chen, Nanotechnology 18, (2007) 205701. <u>https://doi.org/10.1088/0957-4484/18/20/205701</u>
- [9] Y. Hattori, S. Nomura, S. Mukasa, H. Toyota, T. Inoue, T. Kasahara, J. Alloys, Comp. 560 (2013) 105-110. <u>https://doi.org/10.1016/j.jallcom.2013.01.137</u>
- [10] Y.X. Qin, F. Wang, W.J. Shen, M. Hu, J. Alloys Comp. 540 (2012) 21-26. https://doi.org/10.1016/j.jallcom.2012.06.058
- [11] L. Fang, S.J. Baik, K.S. Lim, S.H. Yoo, M.S. Seo, S.J. Kang, J.W. Seo, Appl. Phys. Lett. 96 (2010) 193501. <u>https://doi.org/10.1063/1.3427396</u>
- [12] P.J. Barczuk, A. Krolikowska, A. Lewera, K. Miecznikowski, R. Solarska, J. Augustynski, Electrochim. Acta 104 (2013) 282-288. <u>https://doi.org/10.1016/j.electacta.2013.04.107</u>
- [13] M. Vasilopoulou, L.C. Palilis, D.G. Georgiadou, A.M. Douvas, P. Argitis, S. Kennou, L. Sygellou, G. Papadimitropoulos, I. Kostis, N.A. Stathopoulos, D. Davazoglou, Adv. Funct. Mater. 21 (2011) 1489-1497. <u>https://doi.org/10.1002/adfm.201002171</u>
- [14] J. Meyer, S. Hamwi, S. Schmale, T. Winkler, H.H. Johannes, T. Riedl, W. [29] W.L. Kwong, N. Savvides, C.C. Sorrell, Electrochim. Acta 75 (2012) 371-380. <u>https://doi.org/10.1016/j.electacta.2012.05.019</u>
- [15] S.M. Yong, T. Nikolay, B.T. Ahn, D.K. Kim, J. Alloys Comp. 547 (2013) 113-117. https://doi.org/10.1016/j.jallcom.2012.08.124
- [16] N.S. Ramgir, C.P. Goyal, P.K. Sharma, U.K. Goutam, S. Bhattacharya, N. Datta, M. Kaur, A.K. Debnath, D.K. Aswal, S.K. Gupta, Sens. Actuators B - Chem. 188, (2013) 525-532. https://doi.org/10.1016/j.snb.2013.07.052
- [17] D.S. Lee, K.H. Nam, D.D. Lee, Thin Solid Films 375 (2000) 142-147. https://doi.org/10.1016/S0040-6090(00)01261-X
- [18] D. Manno, A. Serra, M. DiGiulio, G. Micocci, A. Tepore, Thin Solid Films 324, (1998) 44-51.
- https://doi.org/10.1016/S0040-6090(97)01205-4
 [19] M.G. Hutchins, O. Abu-Alkhair, M.M. El-Nahass, K.
- Abd El-Hady, Mater. Chem., Phys. 98 (2006) 401-405. https://doi.org/10.1016/j.matchemphys.2005.09.052
- [20] A. Rothschild, J. Sloan, R. Tenne, J. Am. Chem. Soc. 122 (2000) 5169-5179.
 - https://doi.org/10.1021/ja994118v
- [21] R.S. Vemuri, G. Carbjal-Franco, D.A. Ferrer, M.H. Engelhard, C.V. Ramana, Appl., Surf. Sci. 259 (2012) 172-177.

https://doi.org/10.1016/j.apsusc.2012.07.014

[22] C. Zhang, A. Boudiba, P.D. Marco, R. Snyders, M.G. Olivier, M. Debliquy, Sens. Actuators B - Chem. 181 (2013) 395-401. https://doi.org/10.1016/j.snb.2013.01.082

[23] R. Sivakumar, A. Moses Ezhil Raj, B. Subramanian, M. Jayachandran Trivedi, C. Sanjeeviraja, Mater. Res. Bull. 39 (2004) 1479-1489.

https://doi.org/10.1016/j.materresbull.2004.04.023

[24] Z. Silvester Houweling, John W. Geus, Michiel de Jong, Peter-Paul R.M.L. Harks, Karine H.M. van der Werf, Ruud E.I. Schropp, Mater. Chem. Phys. 131 (2011), 375-386.

https://doi.org/10.1016/j.matchemphys.2011.09.059

[25] K.J. Lethy, D. Beena, R.V. Kumar, V.P.M. Pillai, V. Ganesan, V. Sathe, Appl. Surf. Sci. 254 (2008) 2369-2376.

https://doi.org/10.1016/j.apsusc.2007.09.068

- [26] S. Yamamoto, A. Inouye, M. Yoshikawa, Nucl. Instrum. Methods B 266 (2008), 802-806. https://doi.org/10.1016/j.nimb.2007.12.092
- [27] L.M. Bertus, C. Faure, A. Danine, C. Labrugere, G. Campet, A. Rougier, A. Duta, Mater. Chem. Phys. 140 (2013) 49-59.
 <u>https://doi.org/10.1016/j.matchemphys.2013.02.047</u>
- [28] P.M. Kadam, N.L. Tanwal, P.S. Shinde, S.S. Mali, R.S. Patil, A.K. Bhosale, H.P. Deshmukh, P.S. Patil, J. Alloys Comp. 509 (2011) 1729-1733. https://doi.org/10.1016/j.jallcom.2010.10.024
- [29] W.L. Kwong, N. Savvides, C.C. Sorrell, Electrochim. Acta 75 (2012) 371-380 https://doi.org/10.1016/j.electacta.2012.05.019
- [30] B. Ingham, S.V. Chong, J.L. Tallon, Curr. Appl. Phys. 4 (2004) 202-205. https://doi.org/10.1016/j.cap.2003.11.009
- [31] N. Naseri, H. Kim, W. Choi, A.Z. Moshfegh, Int. J. Hydrogen Energy 38 (2013), 2117-2125. <u>https://doi.org/10.1016/j.ijhydene.2012.11.132</u>
- [32] R. Solarska, B.D. Alexander, A. Braun, R. Jurczakowski, G. Fortunato, M. Stiefel, T. Graule, J. Augustynski, Electrochim. Acta 55 (2010) 7780-7787. https://doi.org/10.1016/j.electacta.2009.12.016
- [33] C.V. Ramana, G. Baghmar, E.J. Rubio, M.J. Hernandez, ACS Appl. Mater. Int. 5, (2013) 4659-4666. <u>https://doi.org/10.1021/am4006258</u>
- [34] H.H. Lu, J. Alloys Comp. 465 (2008) 429-435. https://doi.org/10.1016/j.jallcom.2007.10.105
- [35] S. Keshri, A. Kumar, D. Kabiraj, Thin Solid Films 526 (2012) 50-58.
 https://doi.org/10.1016/j.tsf.2012.10.101
- [36] K.J. Patel, C.J. Panchal, V.A. Kheraj, M.S. Desai, Mater. Chem. Phys. 114 (2009) 475-478. <u>https://doi.org/10.1016/j.matchemphys.2008.09.071</u>
- [37] R. Binions, C. Piccirillo, R.G. Palgrave, I.P. Parkin, Chem. Vapor Depos. 14 (2008) 33-39. <u>https://doi.org/10.1002/cvde.200706641</u>
- [38] A.Z. Mohammed, N.J. Mohammed, I.K. Khudhair, "Effect of the Number Shots of Laser on Structural Transformations and Optical Properties of ZnS

Nanoparticles Thin Films," Arab J. Nucl. Sci. Appl., vol. 51, 4, pp. 108-117, 2018.

[39] Lethy, K. J., Beena, D., Kumar, R. V., Pillai, V. M., Ganesan, V., & Sathe, V. (2008). Structural, optical and morphological studies on laser ablated nanostructured WO3 thin films. Applied Surface Science, 254(8), 2369-2376,

https://doi.org/10.1016/j.apsusc.2007.09.068

- [40] Cullity, B. D. Elements of X-ray Diffraction. Addison-Wesley Publishing, 1956.
- [41] N. J. Mohammed, H. A. Ahmed, "Effect of Laser Fluence on Structural Transformations and Photoluminescence Quenching of Zinc Selenide Nanoparticles Thin Films", Al-Mustansiriyah Journal of Science, Volume 29, Issue 4, PP 122-127, 2018. <u>https://doi.org/10.23851/mjs.v29i4.441</u>
- [42] Corona, S. A. M., Souza, A. E. D., Chinelatti, M. A., Borsatto, M. C., Pécora, J. D., & Palma-Dibb, R. G.' Effect of energy and pulse repetition rate of Er: YAG laser on dentin ablation ability and morphological analysis of the laser-irradiated substrate'. Photomedicine and Laser Therapy, 25(1), 26-33. (2007).

https://doi.org/10.1089/pho.2006.1075

- [43] Shaker, S. S. 'Preparation and Study the Characteristics of Tungsten Trioxide Thin Films for Gas Sensing Application', Engineering and Technology Journal, 34, (2016).
- [44] E. A. Davis, N. F. Mott, Phil. Mag. 22, 903, (1970). <u>https://doi.org/10.1080/14786437008221061</u>
- [45] Díaz-Reyes, J., Castillo-Ojeda, R., Galván-Arellano, M., & Zaca-Moran, O. 'Characterization of WO3 thin films grown on silicon by HFMOD, Advances in Condensed Matter Physics, 2013. https://doi.org/10.1155/2013/591787
- [46] Zou, Y. S., Zhang, Y. C., Lou, D., Wang, H. P., Gu, L., Dong, Y. H., ... & Zeng, H. B.. Structural and optical properties of WO3 films deposited by pulsed laser deposition. Journal of alloys and compounds, 583, 465-470, 2014.

https://doi.org/10.1016/j.jallcom.2013.08.166

How to Cite

A. J. . Hwaidi and P. D. N. J. Mohammed, "Tuning Structural and Optical Properties of WO3 NPs Thin Films by the Fluency of Laser Pulses", *Al-Mustansiriyah Journal of Science*, vol. 33, no. 3, pp. 94–100, Sep. 2022.



100

